137-58-1-1236

Translation from: Referativnyy zhurnal, Metallurgiya, 1958, Nr l, p 166 (USSR)

AUTHORS: Gavranek, V. V., Bol'shutkin, D. N.

TITLE: Surface Har

Surface Hardening as a Means of Improving the Cavitation Resistance of Metals (Povysheniye kavitatsionnoy stoykosti metallov s pomoshch'yu uprochneniya poverkhnosti)

PERIODICAL: Tr. Khar'kovsk. politekhn. in-ta, 1957, Vol 9, pp 169-177

ABSTRACT:

Study of the cavitation resistance of 1Kh13 and EI10 steels to machining of the surface layer has established that electric spark machining of the surface by chromium, stellite, and pobedite does not increase cavitation resistance. The chemical and heat treatment of nitriding substantially increases erosion strength, particularly when the hardened layer consists of the X phase and is of maximum thickness.

M. Sh.

1. Metals-Cavitation-Resistance 2. Metals-Hardening

Card 1/1

S/123/59/000/008/029/043 A004/A002

Translation from: Referativnyy zhurnal, Mashinostroyeniye, 1959, No. 8, p. 113, # 29418

AUTHORS:

Gavranek, V. V., Fuks, M. Ya., Bol'shutkin, D. N

TITLE:

X-Ray Investigation of Cavitation Erosion of Metals

PERIODICAL: Tr. Khar'kovsk. politekhn. in-ta, 1958, Vol. 14, pp. 161-168

TEXT: By X-ray examination, using different methods, the authors investigated the cavitation strength of IX13 grade steel after oil-hardening at 1,000°C and tempering at 680°C. It was found that during the first period of cavitation action a breaking up of the crystals is taking place, while during continued testing the crystal size is stabilized. During the same period, the magnitude of stress of the second kind grows and again returns to its initial value. Those crystallites, located at the surface, are subjected to destruction which are less favorable oriented in relation to the effective stresses from cavitation. The indicated selective destruction is extinguished in the course of time, since the appearing surface relief promotes the destruction of

Card 1/2

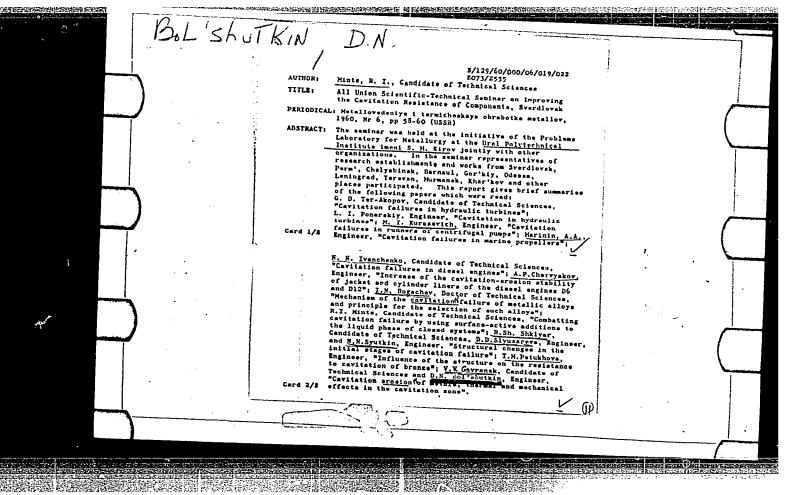
X-Ray Investigation of Cavitation Erosion of Metals

crystallites already independently from orientation. Probably it is possible to increase the erosion resistance in the initial period of destruction by preliminarily producing a favorable structure in the surface layer of metal (e.g. by cold rolling). It is presumed that cavitation destruction by is not taking place owing to plastic deformation, accompanied by a distortion crystallites emerging at the surface are eliminated by way of "breaking off", in the effective zone of impact stresses. There are 4 figures and 8 references.

Translator's note: This is the full translation of the original Russian

Card 2/2

"APPROVED FOR RELEASE: 06/09/2000 CIA-RDP86-00513R000206130013-5



Combination of the first of the state of the

S/126/60/009/05/012/025

AUTHORS: Bol'shutkin, D.N., Gavranek, E021/E335 Fuks, M.Ya.

TITLE: X-ray Investigation of Cavitation Erosion of Metals

PERIODICAL: Fizika metallov it metallovedeniye, 1960, Vol 9, Nr 5,

pp 722 - 725 (USSR)

ABSTRACT: Materials used were the stainless steel 1Kh13 and an

aluminium monocrystal. Cavitation tests were carried out using a magnetostriction vibrator of a frequency of 7.5 kc/s and 0.065 mm amplitude. Each sample was photographed twice by a sharp focusing X-ray camera, focusing the lines (110) and (220). Distribution of the influence of the breaking-up of the crystallites of the mosaic and the size of microstresses, on the diffuse lines, was found by the method of approximation and the method of harmonic analysis. It was shown that cavitation erosion occurred in a similar way to brittle fracture, which has been shown to occur, in the main, by breaking up of crystallites with no substantial microstresses. After

cavitation erosion for up to 20 min, the (110) lines of

the 1Khl3 steel sample showed the presence of an axial texture [110] normal to the surface of the sample. This

X-ray Investigation of Cavitation Erosion of Metals

was produced by preferred disintegration of crystallites with a favourable orientation. Monochromatic and polychromatic radiations were used for the investigation of aluminium. Figure la shows the presence of misorientation of fragments of the monocrystal. Figures 1b and c show that in the initial stages of the test, intensive progressive granulation occurred in individual crystallite fragments. Calculations showed that after 45 sec, the surface was polycrystalline to a depth of 0.15 mm. No substantial microstresses were found. Investigations were also carried out on copper samples. The initial grain size was 150 μ and the grain size of the powder as a result of cavitation erosion was 5 μ . No substantial microdeformation was noted. The results explain the increased cavitation stability of fine-grained alloys. The phase changes occurring in the process of cavitation erosion in the steel U7 were also studied. It was found that tempering of the hardened steel occurred and local temperatures of 450 to 500 °C were reached. There are 4 figures and 4 Soviet references.

Card 2/3

X-ray Investigation of Cavitation Erosion of Metals
ASSOCIATION: Khartkovskie Live Control of Metals

I: Khar'kovskiy politekhnicheskiy institut imeni V.I. Lenina (Khar'kov Polytechnical Institute imeni V.I. Lenin)

SUBMITTED: July 25, 1959

n.b. This paper was presented at the Sixth All-Union

Conference on Applying X-rays for Investigating Materials, June, 1958.

Card 3/3

APPROVED FOR RELEASE: 06/09/2000 CIA-RDP86-00513R000206130013-5"

Bol'ShuTkin, D.N.

5/126/60/010/01/010/019 E111/E335

AUTHORS: Gavranek, V.V., Bol'shutkin, D.N. and Zozulya, V.F. TITLE:

Microfractographic Investigation of the Cavitation Erosion of Metals

PERIODICAL: Fizika metallov i metallovedeniye, 1960, Vol. 10, No. 1, pp. 84 - 89

The authors describe their use of the technique of micro-TEXT: examination of fractured surfaces, previously used by some other investigators (Refs.1-3) in studies of alloy fracture, for getting information on cavitation erosion of metals. Changes in relief of eroded metal after cavitation tests of various durations and the influence of heat treatment and chemical composition on relief structure of alloys were studied. Annealed type 1Kh13 chromium stainless steel, type U7 carbon steel hardened to martensite structure and tempered for 1 hour at 100-600 °C. types Br.A2, BrA4 and Br.A6 aluminium bronzes in the annealed state and types BrA10, BrA12 and Br.A13 in both annealed and hardened states were studied. Cavitation tests were made with a magnetostriction vibrator (Ref. 4) at 75. cps in water. Microexamination of eroded specimens was effected with the aid of

S/126/60/010/01/010/019 **E**111/**E**335

Microfractographic Investigation of the Cavitation Erosion of Metals

titanium replicas (Ref 5). Photographs were obtained with a type EM-3 electron microscope at X1200. The characteristic appearance of brittle-fracture relief type 08KP steel and ductile fracture of chromium-nickel steel are shown in Figure 1 (left and righthand, respectively). Orientation and size of planes was also determined and compared with erosion speed (Ref. 6). Fig. 2 illustrates the surface relief of type 1Kh13 steel in the peripheral and central parts of the specimen and after a 3-minute test and the same after 90 minutes. The reliefs of type U7 steel apecimens after tempering at 100, 400 and 600 $^{\circ}\text{C}$ and cavitation testing for 3 hours are compared in Figure 3 and those of Br.A2. Br.A6 and Br.A13 aluminium bronzes after 3-hours' cavitation testing in Fig. 4. The relief obtained with specimens of hardened Br.A10 and Br.A13 aluminium bronzes after 3-hours testing is shown in Fig. 5. With the alloys studied cavitation erosion occurs by way of brittle fracture of crystals. The size and mutual orientation of planes from which crystals have broken away determine the erosion stability of the alloy: the smaller the planes and the

S/126/60/010/01/010/019 E111/E335

Microfractographic Investigation of the Cavitation Erosion of

degree of their disorientation the greater the stability. Stability can be increased either by hardening and tempering or by additional alloying. There are 5 figures, 1 table and 7 references: 5 Soviet and 2 French.

ASSOCIATION:

Khar'kovskiy politekhnicheskiy institut im.

V.I. Lenina (Khar'kov Polytechnical Institute im.

V.I. Lenin)

SUBMITTED:

September 16, 1959

Card 3/3

18.8300

28978 S/587/59/021/004/002/004 E091/E435

AUTHORS:

Gavranek, V.V. and Bol'shutkin, D.N.

TITLE:

X-ray investigation of cavitation erosion of

monocrystals of aluminium

SOURCE:

Khar kov. Politekhnicheskiy institut. Trudy. v.21, no.4.

1959. Seriya metallurgicheskaya. 17-22

TEXT: It was found by V.V. Gavranek, M. Ya. Fuks and D.N. Bol'shutkin (Ref.1: Fizika metallov i metallovedeniye, 1955, v.1, no.3, 494) that erosion of metals under cavitation conditions is mainly the result of brittle fracture of crystals and that distortions of the crystal lattice caused by stresses of the second type develop only slightly in the initial stages of erosion. further information on the mechanism of cavitation destruction of In order to obtain materials, the authors investigated the structure of metallic monocrystals after they have been subjected to cavitation testing. Monocrystals of commercially pure Al were prepared by means of recrystallization annealing aluminium plates, 200 x 20 x 1 mm^3 , which had first been deformed 3% in tension. carried out at a temperature of 550°C, which was then raised to 590 and 630°C, the specimens being scaked at each temperature for

X-ray investigation of ...

28978 \$/587/59/021/004/002/004 B091/E435

4 hours and subsequently furnace-cooled. monocrystals 70 x 20 x 1 mm3 were obtained. As a result, cavitation testing on a magnetostriction wibrator were cut from the monocrystal plates in the form of dises of 18 mm diameter. were secured to a nickel tube by means of a special holder and subjected to cavitation testing in de-aerated tap water at 25°C for 15, 30, 45, 50, 90, 120 and 300 sec at a tube oscillation frequency of 7500 c/s and constant oscillation amplitude. were taken of the specimens before and after testing. X-ray pictures found that cavitation erosion of monocrystals of aluminium is accompanied by intense break-down of these crystals into fragments, so that already after 45 see the surface of the specimens becomes polycrystalline to a depth of 0.15 mm, the grain size being 10-4 cm. It was also found that the cavitation destruction of Al monocrystals does not cause great lattice distortions of the second stress type. The authors express the opinion that the brittle destruction of metals under conditions of cavitation erosion may be due to propagation of stress waves created by the impact action of cavitation bubbles. There are 6 figures and 5 references: 4 Soviet and 1 Russian translation of an English book.

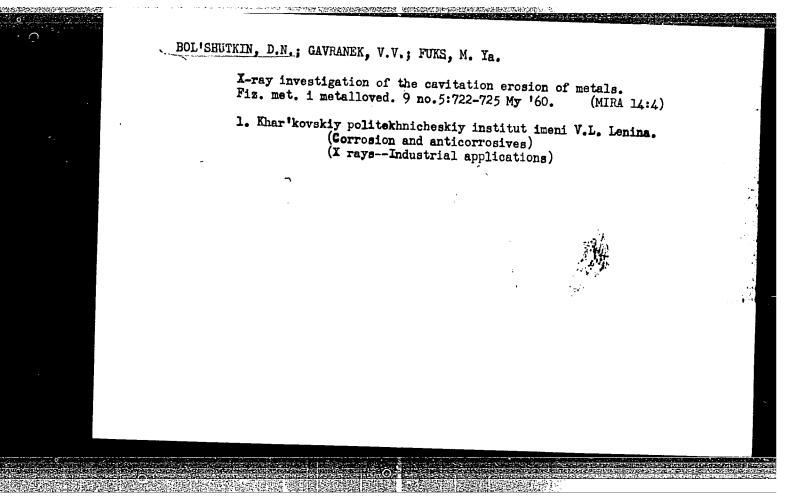
X-ray investigation of ...

28978 \$/587/59/031/004/002/004 B091/E435

reference to the English language publication meads as follows: Ref.3: Barret, C.S., Structure of Metals. Russian translation Metallurgizdat, 1948.

K

Card 3/3



BolsHutkin, DN.

826112

18.8200

5/126/60/010/02/012/020 E021/E335

AUTHORS:

Gavranek, V.V., Bol'shutkin, D.N. and Zel'dovich, V.I.

TITLE:

Thermal and Mechanical Action of a Cavitation Zone

on the Surface of a Metal

PERIODICAL:

Fizika metallov i metallovedeniye, 1960, Vol. 10,

No. 2, pp 262 - 268

The present work is an attempt to examine the change in temperature and pressure impulses arising in the surface layers of a solid in the cavitation zone. A magnetostriction vibrator was used in the experiments with a constant amplitude of $0.06\ \mathrm{mm}$ and a frequency of 7.5 kc/s. Phase changes in a quenched U7 steel and Dl duralumin were investigated in the cavitation zone by microhardness and X-ray investigations. Fig. 2 shows the relation of microhardness with time of cavitation erosion. Fig. 2a is for the steel and 2b for duralumin. The changes in hardness show that the temperature of micro-volumes during cavitation erosion reaches 470 °C. Fig. 3 shows the effect of a preliminary tempering treatment at various temperatures on hardness (Curve 2) and rate of cosion (Curve 1). The rate of erosion is practically unchanged by preliminary heat treatments Card 1/3

82642 \$/126/60/010/02/012/020 E021/E335

Thermal and Mechanical Action of a Cavitation Zone on the Surface of a Metal

up to 400 °C. This shows that the damage occurs on microvolumes of metal, the temperature of which is up to 400 °C. X-ray investigations show that the internal stresses arising in the steel in the process of quenching are removed by cavitation erosion. Similar results were obtained for duralumin. During the experiments, the duralumin became artificially aged, showing that high temperatures are reached during cavitation erosion.

The obtained results can be summarised thus:

1) the mechanical and the thermal effects in the cavitation zone produced by the magnetostriction vibrator were calculated. It was found that in the case of using/7.5 kg/s vibrator, the pressure increases periodically to 550 kg/cm during a period

of about 10^{-5} sec and acts on an area of about 10^{-5} mm². The micro-volumes of the metal bordering on the cavitation bubble are heated to 300 - 500 °C.

2) It was established that during the process of cavitation erosion, hardened steel is being tempered at temperatures up Card 2/3

82642

S/126/60/010/02/012/020 E021/E335

Thermal and Mechanical Action of a Cavitation Zone on the Surface of a Metal

to 470 °C.

3) The speeds of cavitation erosion of steel hardened to obtain a martensitic structure and of steel tempered at temperatures below $400\,^{\circ}\text{C}$ are practically equal. There are 4 figures and 10 Soviet references.

ASSOCIATION:

Khar'kovskiy politekhnicheskiy institut im. V.I. Lenina (Khar'kov Polytechnical Institute im. V.I. Lenin

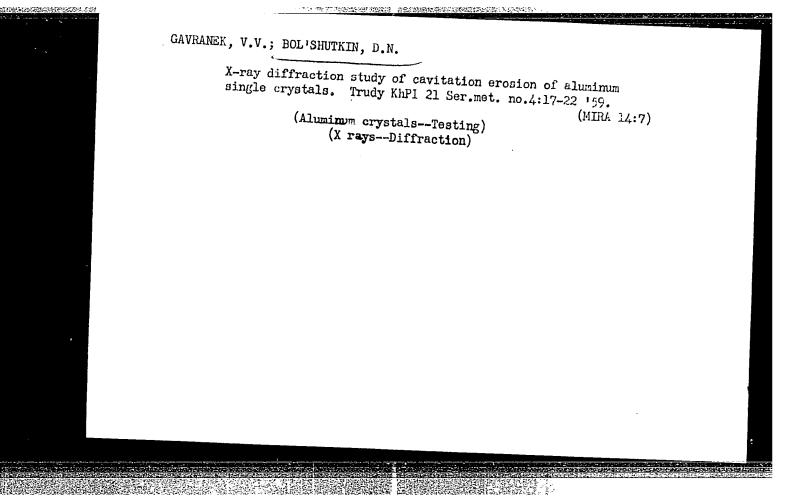
SUBMITTED:

June 26, 1959, originally, February 17, 1960, after revision.

Card 3/3

Sporsocus Sporsocus Sporsocus Sporsocus Scholing Pravile Profession of Technical Card 1/0 Card 1/0 Card 1/0 Card 1/0 Card 1/0 Covenate Pression Pre	Inserted, 5,000 copies printed. Sponsoving Agency: Genderstvenryy nauchno-technichentyk kemitet Seven Ministrov Ukr53. Nauchno-technichentscyc obhinhertvo mashinostroitel'noy prezyshlennosti. Klyvvskoyr oblasinoye pravleniye. Editorial Board: M. P. Brauh, Dostor of Technical Selences, I. Ya. Dokhyar, Doctor of Technical Selences, D. A. Draygor, Dostor of Technical Selences, I. S. Kamentehnyye, Engineer, Yo. A. Parkov of Technical Selences, I. S. Kamentehnyye, Engineer, Yo. A. Parkov of Technical Selences, Mand A. V. Ghennool, Confedete of Technical Selences, Soroka; Tech. Misal Selences; Ed. M. S. Soroka; Tech. M. S. Gonnostapol's Biaya; Cafef Ed., Mashgiz (Southern Dept.): V. K. Gard 1/10	PURCESE That collection of articles is intended for esterific and schools of higher technical education. OVENCES: The collection contains papers presented at a convention their in Kyley on problems of privates metaliners and mithods of the heat treatment of metals applied in the metaline industry. The collection metals and alloys are discussed, and reaults of investigations conducted to ascentin the effect of investigations conducted to ascentin the effect of investigations conducted to ascentin the effect of ability of obtaining metals with kinnets of transformation that treatment, and properties of setal brittlenous. The collection disturdes papers dealing with kinnets of transformation, heat treatment, and properties of setal brittlenous. The collection thinks and properties of eact iron. We presented a series mentioned. Articles are accompanied by references, mostly frame or containing a solute. Streellin, A. I., Engineer, and L. A. Melinikov (Sycridovsk). Transformation of Austenite Into Fartenaite Under High Pressure in the Pressure of the Decomposition Kinetics of Partenit, in Respiration of the Decomposition Kinetics of Partenit, in Kinerinite and Iron-Carbon Alloys Mirovelly, E. I., Engineer (Kiyev). The Mature of the phase Truns formation of Garbon Steeln and Tunns formation of Garbon Steeln and Alloys
--	---	--

	6,21d 6,A0
157	, Engineer, and D. H. Bolighykha (Kharikov). Sevitation Erecijon of Metuln
152	-
146	, Engineer (Kiyev). Plakeo in Steel
138	Rarmov, S. M. (Leningrad). Hydrogen as a Surface-Active la Admixture in Alloys
132	Braynin, I. Ye., Doctor of Technical Science, Frofersor (Stalino), V. A. Murchenko, Engineer and A. I. Kondinshev (Kramatoruk). Experimental Investigation of Streen Distribution in the Gross Section of a Bent Billet as Related to Flaking.
121	
	Physical Natallurgy (Cont.)
	Vinokur, B. B., Engineer, E. I. Mirovakiy (Kiyev) and A. L. Geller (Kramaterak). Effect of the Increase of Forging
121	Sokol, A. N., Candidate of Technical Sciences, O. S. Kostyrko, Engineer, E. I. Mirovskiy, B. B. Vinkin, and K. P. Braun, Doctor of Archical Sciences, Professor (Kiyev), Plasticity of Steels dithin the Pressvorking Temporature Pange
114	
106	
16	÷4
	Physical Notallurgy (Cont.)
92	Barunov, S. M., Doctor of Technical Sciences, Professor (Leningrad). Effect of Silicen Monoxide on the Properties of Steel
ıΛ	dolovan', N. A., Engineer (Kijer). Concentration of Carbides 85
10	Cherepin, V. T., Candidate of Technical Sciences (Kiyev). 75 Terpering of Carbon Steel by Using Electric Heating
N	Pernyakov, V. G., Engineer, and M. V. Beloua (Kiyev). The Changes in the Carben Brase During the Tentralia of Carben, 62 Silicen, and Aluminum Steels
•	Sidovakly, V. D., Engineer, and G. M. Eogacheva. (Sverilovan). Gr. the Prolem of the Inace Recrystallization of the Glad- Cast Steel
	Rauzin, Yz. H., Engineer (Moseex). On the Nature of the 44 Critical Degree of Strain
•	



GAVRANEK, V.V., kand tekhn.nauk, dotsent; BOL'SHUTKIN, D.N., kand tekhr.reft;

VOLONTSEVICH, O.A., inzh.

Investigating the erosion strength of steel hardened by electric spark treatment and subjected to grinding. Vest.mashinostr. 43 no.9:62-64 S '63. (MIRA 16:10)

BOLISHOTKIN, P.N.; ISONTIYEVA, A.V.; SNIGIREV, V.C.; STARTSEV, V.I.

Fragnoss of crystelline methane. Fiz. tver. tela 7 no.9:2607-2611 S
(MTRA 18:10)

1. Fiziko-tekhnicheskiy institut nizkikh temperatur AN Ukr9SR,

khorikov.

ACC NR AP5022724 SOURCE CODE: UR/0181/65/007/009/2789/2792 44,55 411,55 AUTHOR: Bol'shutkin, D. N.; 40,55 Prokhvatilov, A. I.; Sil'vestrova, T. V.; Startsev, 44,55 ORG: Physicotechnical Institute of Low Temperatures AN UkrSSR, Kharkov (Fiziko tekhnicheskiy institut nizkikh temperatur AN UkrSSR) TITLE: Mechanical properties of polycrystalline ammonia under unilateral compres-SOURCE: Fizika tverdogo tela, v. 7, no. 9, 1965, 2789-2792 TOPIC TAGS: ammonia, solid mechanical property, low temperature physics ABSTRACT: The strength and ductility of polycrystalline ammonia are studied as functions of temperature under unilateral pressure. Cylindrical specimens 10 mm in diameter and 40 mm long with uniform microstructure and polished ends were studied at temperatures from 77 to 160°K. Curves are given for the breaking point, limit of proportionality and relative compression as functions of temperature. These data show that crystalline ammonia has extremely low strength properties and ductility. Solid ammonia is quite brittle at the temperature of liquid nitrogen and shows elastic deformation right up to the breaking point. At stresses of 0.5-0.6 kg/mm²,cracks are formed parallel to the axis of the specimen with an accompanying characteristic Card 1/2

sound and a slight reduction in loading (up to 100 g). The final breaking stress of 0.8 kg/mm^2 remains constant throughout the experimental temperature range. At this point there is an instantaneous reduction in loading to zero and the specimen is shattered. The shape of the fragments and the slight degree of deformation before the breaking point show that cleavage is the mechanism responsible for fracture of ammonia crystals between 77 and 130°K . Above 130°K (0.6 T), the ductility of the specimens increases and creep is observed under a constant load. Shearing is responsible for fracture above this point since cleavage strength remains nearly constant with temperature, while an increase in temperature causes a considerable reduction in shearing strength. The relationship between rate of uniform creep V and stress of is $V = A\sigma^n$, where A and n are constants equal to 500 and 5 respectively at 160°K and stresses greater than the limit of proportionality. The energy of creep activation is found to be 5.6 Kcal/mol. This is approximately 10°k lower than the heat of sublimation for solid ammonia. Orig. art. has: 3 figures.

SUB CODE: 07,20/ SUBM DATE: 16Apr65/ ORIG REF: 005/ OTH REF: 010

Card 2/2 DW

ACC NR: AP5022724

L 26630-66 EWr(m)/EWP(w)/T/EWP(t) JD/JG ACC NR: AP5025376 SOURCE GODE: UR/0126/65/020/003/0465/0467 AUTHOR: Bol'shutkin, D. N.; Krot, Yu. Ye.; Moskalenko, V. A. کامی B ORG: Physico-Technical Low Temperature Institute AN USSR (Fiziko-tekhnicheskiy institut nizkikh temperatur AN USSR) Study of lanthanum and neodymium hardness as a function of temperature between 770K and 2930K SOURCE: Fizika metallov i metallovedeniye, v. 20, no. 3, 1965, 465-467 TOPIC TAGS: Lanthanum, neodymium, hardness, temperature dependence, cryogenic effect, phase transition, liquid nitrogen, induction furnace, vacuum furnace ABSTRACT: The system studied consisted of 99.3% lanchanum containing 0.3% Ce; 0.1% Nd; 0.2% Pr; 0.02% Fe, and neodymium containing 99.2% neodymium and < 0.5% Pr, < 0.1% Sm, < 0.002% Ca, < 0.05% Fe. Samples were prepared in a vacuum induction furnace. Measurements of hardness were made by means of Vikker's apparatus equipped with a low temperature modification. Liquid nitrogen was used to obtain temperatures in the range of 77-293°K. A heater was attached for the evaporation of liquid nitrogen. It was found that 40% deformation at room temperature increased the hardness of both metals by 60% as compared to the Card 1/2 UDC: 620.178.15

L 26630-66	
ACC NR: AP5025336	701
heated samples. Superimposed curves of lanthanum and neodymium show analogous temperature dependence for both metals. By cooling these metals from 293 to 77°K their hardness increased by 50% for heat-treated samples and by about 40% for cold-worked samples. The absence of polymorphic transformations of lanthan in the temperature range of 77°-293°K, regardless of its close resemblance to 3 figs.	
$\{ v_i \mid v_i \in H(x_0) : v_i \in V(x_0) \in X(x_0) \mid v_i \in V(x_0) \}$	
SUB CODE:11,20,15 BM DATE: 140ct64/ ORIG REF: 002/ OTH REF: 002	
 Card 2/2 //	

L 21396-66 EVIT(m)/EVP(t)

IJP(c) MC/WW/QC

ACC NR: AP6003800

SOURCE CODE: UR/0181/66/003/001/0248/0249

AUTHORS: Bol'shutkin, D. N.; Prokhvatilov, A. I.

ORG: Physicotechnical Institute of Low Temperatures AN UkrSSR Khar's kov (Fiziko-tekhnicheskiy institut nizkikh temperatur AN UkrSSR)

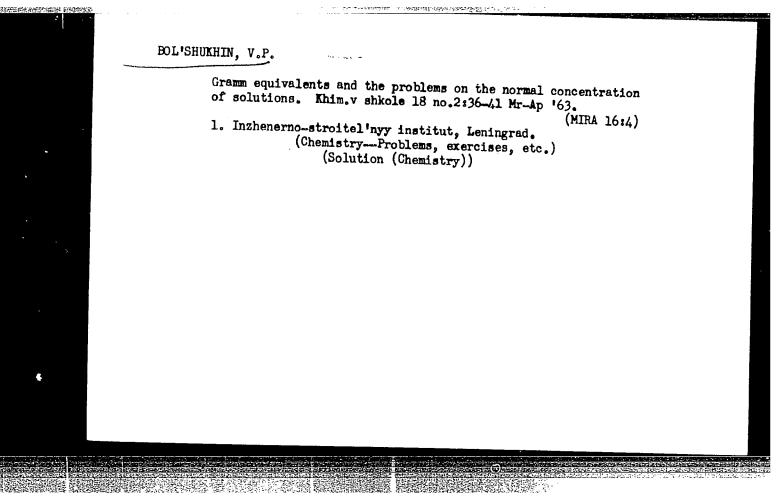
Temperature dependence of the moduli of elasticity of crystalline ammonia

Fizika tverdogo tela, v. 8, no. 1, 1966, 248-249 SOURCE:

TOPIC TAGS: ammonia, temperature dependence, elastic modulus, elastic deformation, creep mechanism, stress relaxation

ABSTRACT: The authors used earlier compression-test data (FTT v. 7, 2789, 1965) to determine the static modulus of normal elasticity of polycrystailine ammonia in the temperature interval 77 -- 160K. Inasmuch as at T > 110K a great reduction is observed in the limit of proportionality of ammonia, making measurements on the deformation curve difficult, they made use of the fact that the proportionality limit can be increased by prior deformation of the samples. Conse-

21396-66 ACC NR: AP6003800 quently the values of the modulus of normal elasticity were determined from the second-deformation curves the accuracy was 7 per cent. A plot of the temperature variation of the modulus of elasticity shows that the values of the dynamic modulus of normal elasticity differ little from the static modulus determined by the static method. With increasing temperature, the difference between the two increases from 14 per cent at 77K to 21 per cent at 160K, owing to the increased role of the relaxation processes and creep processes upon deformation of the crystalline ammonia. From the shear modulus of the crystalline ammonia, it is estimated that its theoretical strength is ~10 kg/mm². By comparison with the earlier results, it is concluded that the strength of the investigated polycrystalline ammonia was 15 times lower than theoretical. Orig. art. has: 1 figure and 2 formulas. SUB CODE: 20/ SUBM DATE: 21Ju165/ ORIG REF: 002/ OTH REF: 003



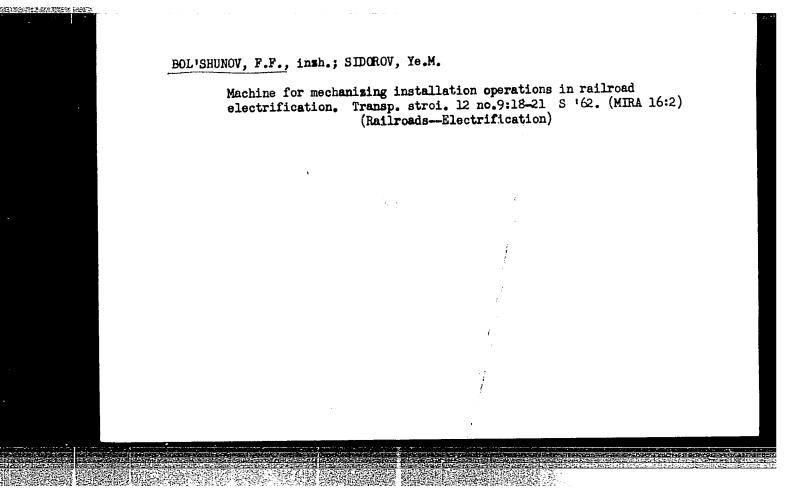
ISAKOV, P.P., SHAHYTIH, L.I., SHCHERBAKOV, V.A.; MAKARENKO, V.I.;

BOLI-SHUKHIN, V.S.; PIVHIK, M.M., CHUDAKOV, V.D.; YAKOVLEV,

G.S.;

[DET--250 diesel-electric tractor; its construction and operation] Dizel'-elektricheskii traktor DET-250; ustroistvo i ekspluatatsiia. Moskva, Mashinostroenie, 1965. 479 p.

(MIRA 18:7)

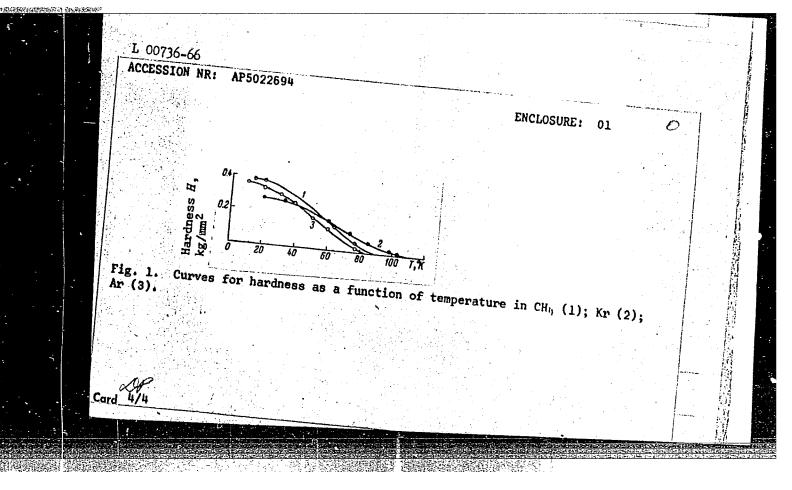


L 00736-66 EPF(c)/EWT(m) ACCESSION NR: AP5022694 AUTHOR: Bol'shutkin, D. N.; Leont'yeva, A. TITLE: Hardness of crystalline methane SOURCE: Fizika tverdogo tela, v. 7, no. 9, 1965, 2607-2611 TOPIC TAGS: hardness, methane, solid state ABSTRACT: The authors study the effect of temperature on hardness and creep in polycrystalline methane at nitrogen and hydrogen temperatures. Since methane is actively dissolved by nitrogen and hydrogen, the specimens were prepared and their hardness was determined in the same hermetically sealed cryostat. The specimens were transparent without visible flaws and had a smooth horizontal surface. The hardness of the methane was determined by sinking a conical indenter into the specimen. The loading unit of the instrument consists of a metal cylinder with a weight of P = 600 g. On the lower section of the cylinder are three conical indenters with vertex angles of 90° located equidistantly around the cylinder. Penetration of the indenter was monitored on a cathetrometer with an accuracy of 0.01 mm. In determining the hardness, penetration of the indenter is given as $h = h_1 + h_2$ where h_1 is penetration under a load P_1 = 10 g, which is read on a spring indicator; h_2 Card 1/4

	L 00736-66	
	ACCESSION NR: AP5022694	
	is penetration of the indenter when the load is increased from 10 to 200 g. Since	
	and consequently $H = \frac{P}{\pi h^2} = \frac{P_1}{\pi h_1^2}$	
	$H = \frac{P}{\pi \left(\sqrt{\frac{P_1}{\pi H} + h_2} \right)^2},$ from which the first state of the first stat	
	from which the following relationship was derived for calculating the hardness $H = \frac{(\sqrt{P} - \sqrt{P_1})^2}{\pi h_2^2}.$ The hardness	
t	of the Enclosure. Curves for argon and krypton are given for comparison (C. Trepp, or reduce the creep effect in solid methane. A physical explanation is given for he effect of temperature on hardness and creep on the basis of the dialog.	
r.co	ord_2/4	

	ACCESSION NR: AP5022694		
	theory. "In conclusion, the authory work and valuable advice, and V. 2 figures, 8 formulas, 1 table. ASSOCIATION: Fiziko-tekhnicheskiy (Physicotechnical Institute of Low		orig. art. has: 4
	SUBMITTED: 15Feb65 NO REF SOV: 003	ENCL: 01 OTHER: 005	SUB CODE: SS
200	Card 3/4		

"APPROVED FOR RELEASE: 06/09/2000 CIA-RDP86-00513R000206130013-5



BOGUSH, A.A.; BOLSUN, A.I.

General calculation of the matrix elements for polarized vector particles. Dokl. AN SSSR 155 no. 5:1046-1049 Ap '64.

(MIRA 17:5)

1. Institut fiziki AN SSSR. Predstavleno akademikom V.A. Fokom.

L 45656-65 EWT(m)/EWA(m)-2 Peb ACCESSION NR: AP5009832

UR/0367/65/001/002/0288/0290

AUTHOR: Bogush, A. A.; Bolsun, A. I.; Satsunkevich, I. S.

19

TITLE: On the scattering of antineutrinos by electrons

B

SOURCE: Yadernzya fizika, v. 1, no. 2, 1965, 288-290

TOPIC TAGS: antineutrino scattering, electron scattering, form factor, vector meson, charge distribution, nonlocality parameter

ABSTRACT: The differential and total cross sections for the process $v_c + e \rightarrow e' + e'$, which has hitherto been discussed from the point of view of a weak contact interaction by several authors, are calculated in this paper by assuming the existence of intermediate vector mesons and by taking into account the possible charge distribution of the antineutrino. The possible information that can be obtained with existing experimental techniques is discussed, and possible methods of determining the nonlocality parameter due to the bosons and the Dirac charge form factor of the antineutrino are described. It is concluded that an account of the antineutrino charge form factor may be very important in the interpretation of experimental data on the process in question, and that if this form factor is small

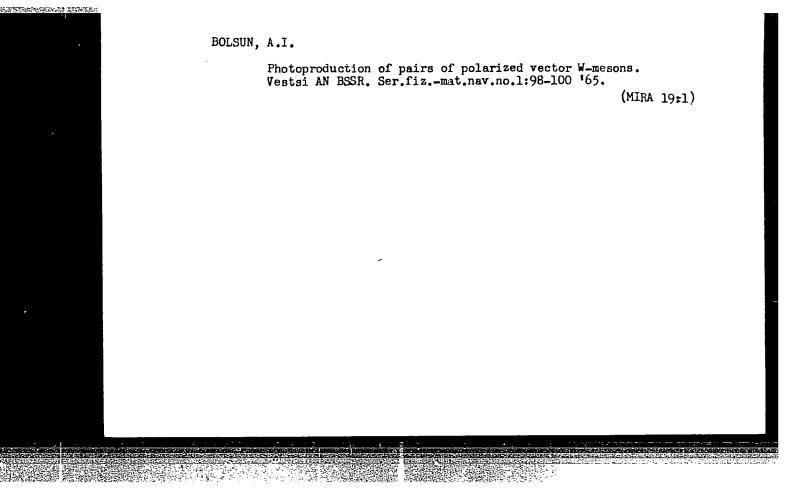
Card 1/2

and the neutral vector meson does indeed exist, then the process may not be observable at all. "The authors thank F. I. Fedorov and L. G. Moroz for useful comments." Orig. art. has: 1 figure and 8 formulas. ASSOCIATION: Mosk-vskiy inzhenerno-fizicheskiy institut (Moscow Engineering						
Physics institute) SUBMITTED: 01Ju164	ENCL: 00	Sub Code: NP				
NR REF SOV: 003	OTHER: 007					
ML Card 2/2						

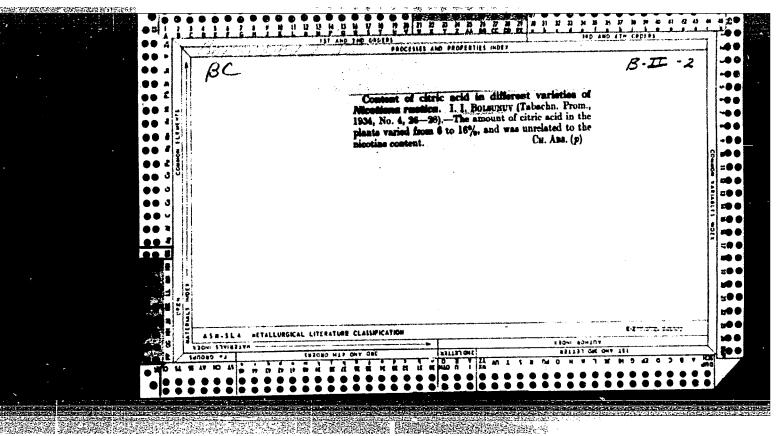
EOLEUN, A.1.; SATSUPKEVICH, I.S.

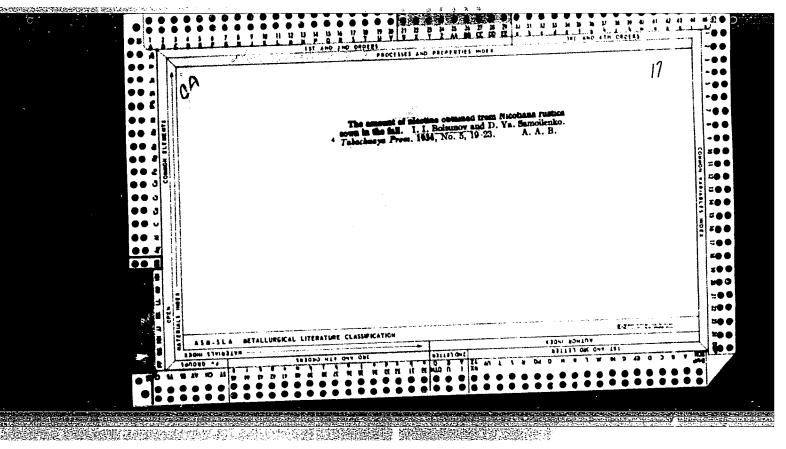
Photoproduction of intermediate vector mesons on protons. Dokl.
AN BSSR 8 no.11:705.708 N '64. (MIRA 18:3)

1. Institut fiziki AN BSSR.



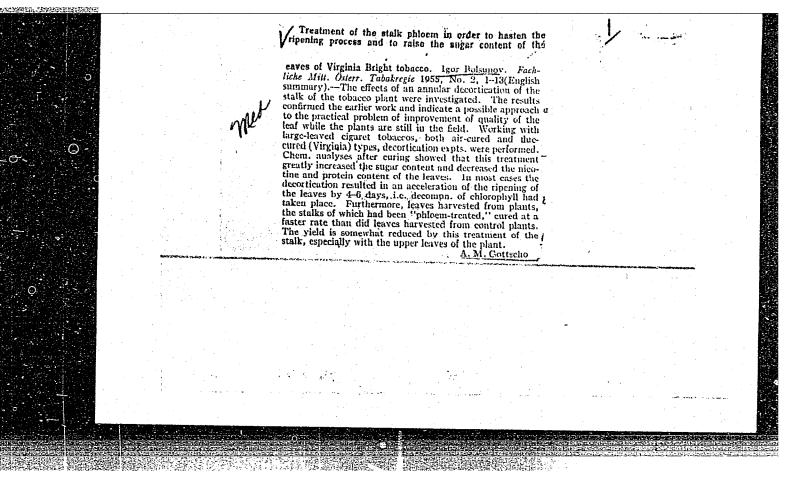
I 100 3 3-1	6 日本(元 <u>)</u>		The state of the s						
	AP6030371	SOURCE CODE:	UR/0428/66/000/001/0099/0106						
• AUTUIN	Bolsun, A. I.; Fedorov, F. I.		25- -3						
AU MOR.	DOISHI, A. I., Federov, 1. I.		ے ۔						
ORG: no	ne		19						
TITLE:	Pseudoscalar matrix beta sub 5 and	the electrical	/ 1						
SOURCE:	AN BSSR. Vestsi. Seryya fizika-n	natematychnykh	navuk, no. 1, 1966, 99-106						
TOPIC TA	GS: meson, dipole moment								
ABSTRACT: This paper, which was discussed at a seminar at the Theoretical Physics Laboratory, Institute of Physics, Academy of Sciences BSSR, shows that the \mathcal{A}_5 matrix is necessary for the introduction of pseudovector interactions related to the internal electrical dipole moment (EDM) of a W-meson. The matrix and its properties, the EDM of the W-meson, and the effect of the latter on the processes e + e W + W are discussed in detail. The authors thank all who participated in the seminar at the Theoretical Physics Laboratory, Institute of Physics, BSSR, for the valuable discussions. Orig. art. has: 1 figure and 35 formulas. [JPRS: 35,668]									
SUB CODE	E: 20 / SUBM DATE: 20Dec65 / 6	ORIG REF: 007	/ OTH REF: OlO						
Card 1/1	MILP		0918 1094						





#Selection of Micotiana rustica and its progress" (p. 336) by Bolsunov, I. I.

**So: Advances in Modern Biology* (Uspekhi Sovremennoi Biologie) Vol. XII, No. 2, 1940



PRUTSKOVA, M.G., kand. sel'khoz. nauk; UKHANOVA, O.I.; SAKHAROVA, L.I.;

BOLSUNOVSKYA, O.V.; IVAKOVA, N.Ye.; LOVCHIKOV, I.S.; ZALKIND,

G.N.; IL'IN, M.I.; KOZ'MINA, K.A.; SHIKUT', V.A.; PETROVA,

Z.V.; GENERALOV, G.F.; BUDIUK, V.P.; GOMENYUK, L.I., red.

[New highly productive varieties of grain crops] Novye'vysokoproduktivnye sorta zernovykh kul'tur. Moskva, Kolos, 1965.

319 p. (MIRA 18:8)

PRUTSKOVA, M.G., kand. sel'khoz. nauk; BOLSUNOVSKAYA, O.V., agronom; LOVCHIKOV, I.S., agronom; MARINICH, P.Ye., red.; KONDRATOVA, N.A., red.; PECHENKIN, I.V., tekhn. red.

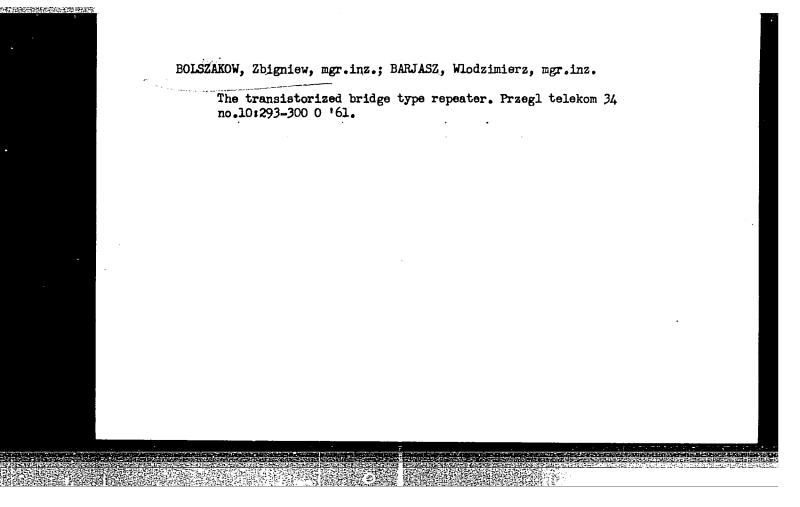
[New strong and durum spring wheat varieties; Saratov 29, Saratov 210, Bezenchuk 98, Kharkov 46, Melianopus 26] Novye sorta sil'nykh i tverdykh iarovykh pshenits; Saratovskaia 29, Saratovskaia 210, Bezenchukskaia 98, Khar'kovskaia 46, Melianopus 26. Moskva, Izd-vo M-va sel'.khoz. SSSR, 1960. (MIRA 14:8)

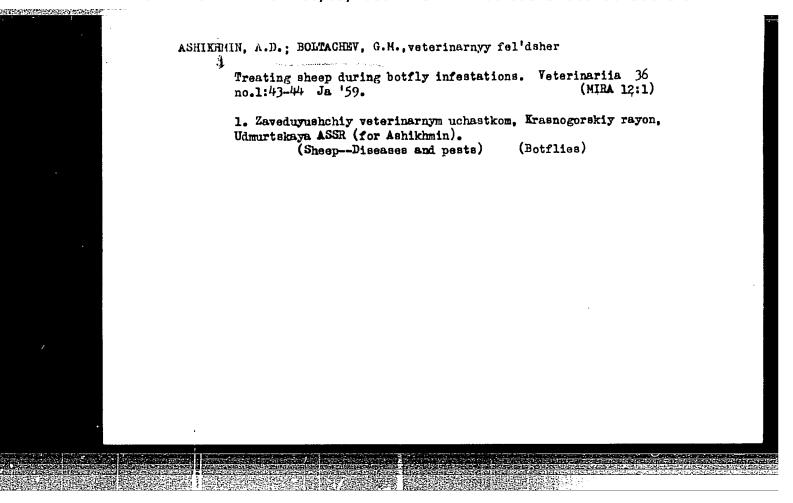
1. Russia(1923- U.S.S.R.) Gosudarstvennaya komissiya po sortoispytaniyu sel'skokhozyaystvennykh kul'tur. 2. Zamestitel' predsedatelya Gosudarstvennoy komissii po sortoispytaniyu sel'skokhozyaystvennykh kul'tur (Marinich) (Wheat--Varieties)

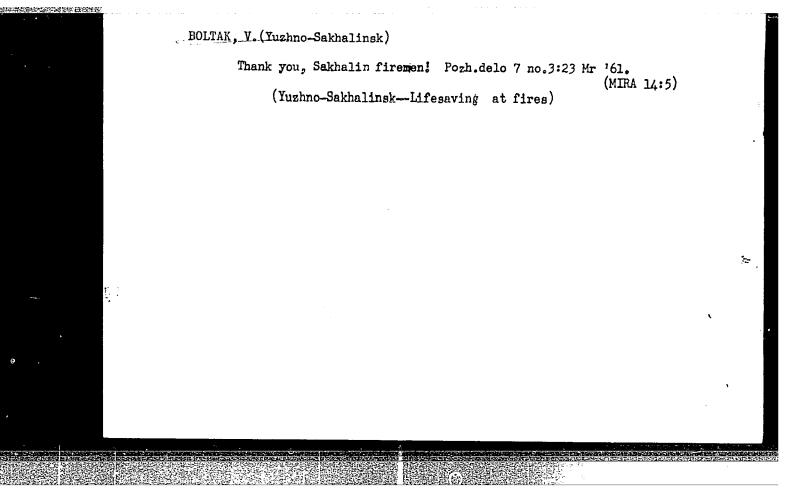
BOLSUNOVSKIY, L.I.

Using the laboratory outfit for radio engineering. Fiz. v shkole 20 no.5:82-83 S-0 160. (MIRA 13:11)

1. Pedagogicheskiy institut, g. Kremenets. (Radio)





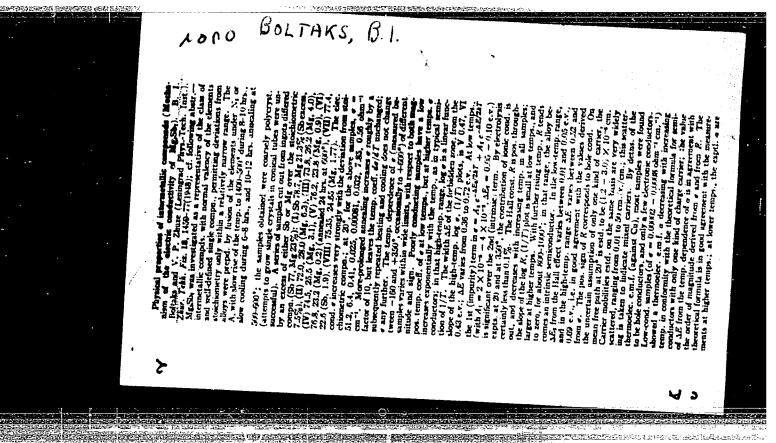


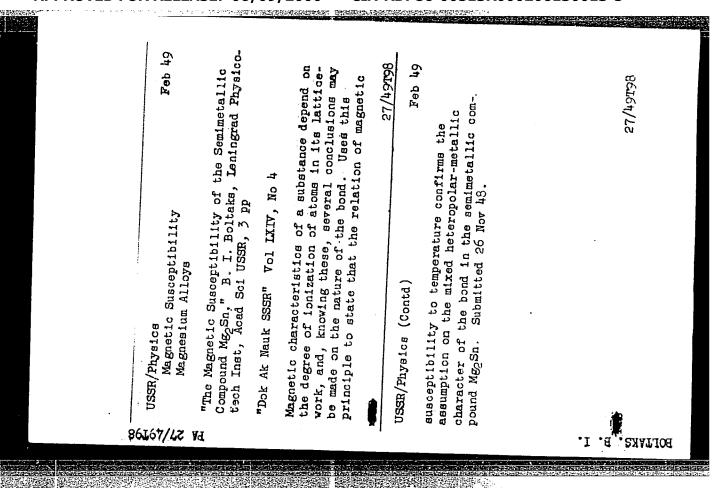
BOLTAKS, A. I.

Organization of practical laboratory work in the measurement of certain meteorological elements. Meteor.i gidrol. no.5:60-61 My '53. (MLRA 8:9)

1. Vyssheye arkticheskoye morskoye uchilishche im. admirala S.O.Makarova, Leningrad.

(Meteorology)





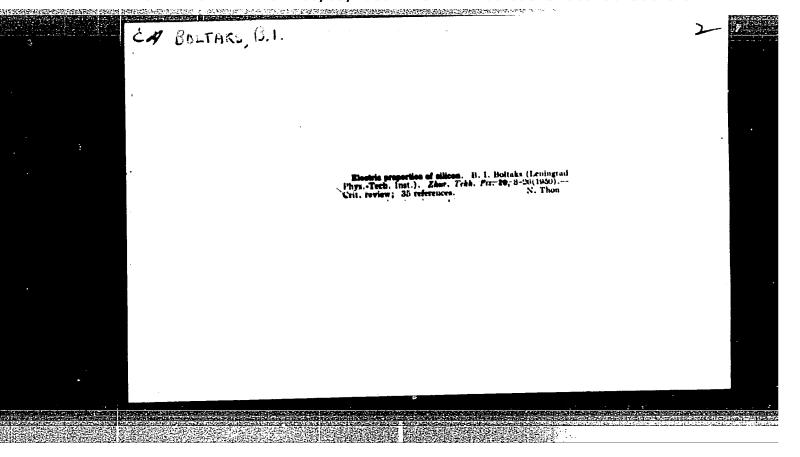
BOLTAKS, B. I.

USER/Chemistry - Magnesium Alloys, Tin Feb 49
Chemistry - Electroconductivity

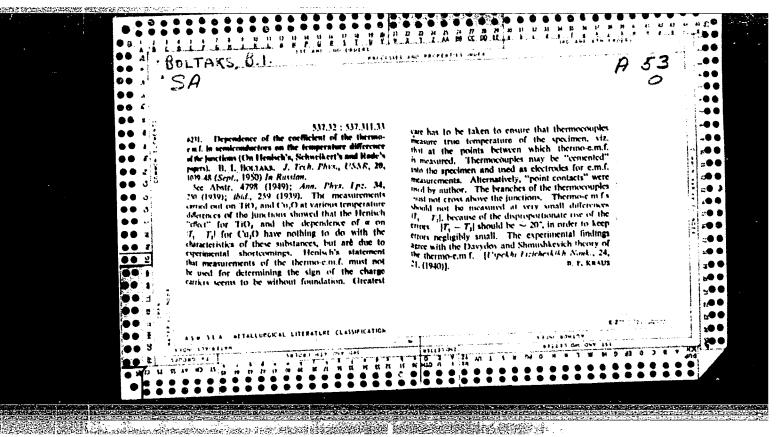
"Electrical Character of Intermetallic Bond of MgpSn,"
B. I. Boltaks, 4 pp

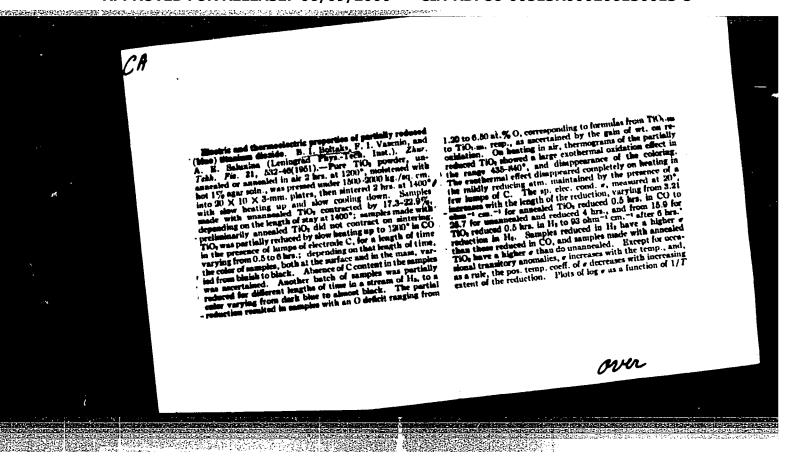
"Dok Ak Nauk SSSR" Vol LXIV, No 5

Attempts to supplement material available on electroconductivity of MgpSn by presenting data on dependencies of electroconductivity, thermoelectromotive force, and Hall's effect upon temperature. Submitted by Acad A. F. Loffe, 26 Nov 48.



Feb 50 properties studied, concludes Mg2Sn must belong 156T103 Feb 50 156T103 Studies dependence upon temperature of specific on temperture behavior of magnetic susceptibilelectrical conductivity, Hall constant, thermoto class of admixed semiconductors. From data ity, makes conclusion on composite (mixed) heemf and magnetic susceptibility of intermetal-"Nature of the Electrical Properties and Magteropolar and metallic character of bond in lic compound Mg_Sn. On basis of electrical netic Susceptibility of the Intermetallic Compound Mg2Sn," B. I. Boltaks, Leningrad Physicotech Inst, Acad Sci USSR, 13 pp USSR/Physics - Magnetic Susceptibility USSR/Physics - Magnetic Susceptibility Intermetallic Compounds Submitted 21 Jul 49. "Zhur Tekh Fiz" Vol XX, No 2 (Contd) this compound. 156T103 Αđ • B ECLTAKS,

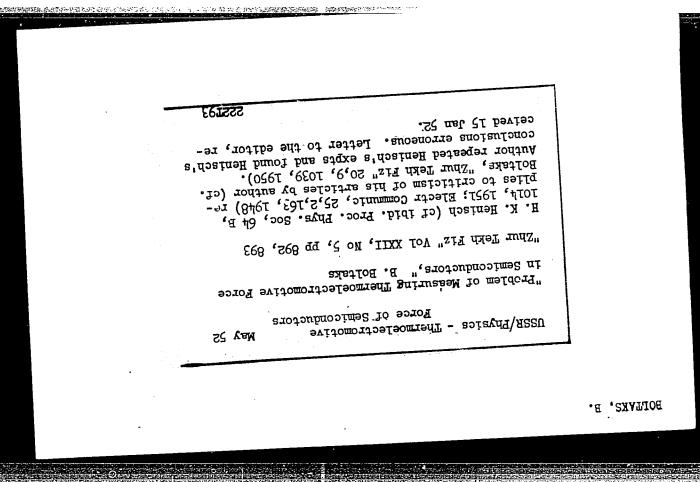


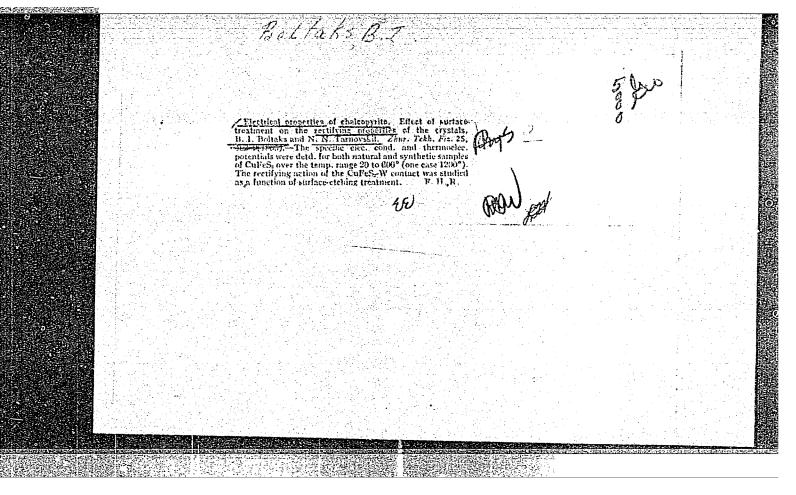


are not linear, i.e. the preexponential factor A in σ on $A_{IC}^{-}\Delta E_{I}/2\hbar T$ is not temp,-independent; linearity is restored if the temp, dependence of A is taken into account by σ on $A_{IC}^{-}\Delta E_{I}/2\hbar T$. By the slopes of the Langusts, the electron dissors, energy ΔE_{I} , around room temp,, is less than 0.2 e.v.; ΔE_{I} increases with increasing A_{I} . The thermoelec, e.m.f. σ (measured against Cu) is neg, throughout, i.e. all the TiO_I samples are electronic conductors. Around room temp,, a decreases with increasing length of reduction; for samples reduced in CO, a varies from 415 microv-/degregion 0.5 hr, to 254 for 6 hrs., and for samples reduced in H_{II}, from 203 for 0.5 hr, to 254 for 6 hrs., with increasing length of reduction, the temp, coeff. of a passes from neg. (typical for semiconductors) to pos., which corresponds to the conce. a of electrons increasing with the temp, slower than TV_I. No Hall effect could be detected; judging by the semistivity of the app, the Hall const. must be smaller than 2 × 10⁻² cc./coulomb, which corresponds to a greater than (3-4) × 10²⁰/cc., and a mobility a of the order of $10^{-2} = 10^{-1}$. cm. 3^{\prime} /v. sec. If the effective mass m^{\prime} of the electrons in partially reduced TiO, were identical with the mass m_{i} of the receiver, a (calcd. from the exptl. a) would be 5 × 10^{10} /cc., and, with a of the order of 10^{-100} cm. 3^{\prime} /v. sec., σ would be of the order of 10^{-5} (10) ohm $^{-1}$ cm. $^{-1}$ (in agreement with the exptl. data, but then the Hall effect should be casily detectable. This discrepancy indicates a substantially greater n. and, consequently, on account of the

high α_i , the effective mass m^a must of necessity be substantially greater than m_i ; this, in turn, corresponds to smaller n_i which accounts for the relatively low a_i . The statio m^a/m_a can be estd. roughly with the aid of a model in which the 2 electrons around the pos. vacancy (produced by the absence of O^{-a} in a lattice point) are treated as a Helike atom, with the ionization energy $E = (24 \times 48/a^2)(m^a/m_a)e.v., where <math>a_i = \text{dielec. const.}$; E is identified with $\Delta B = \text{dissocn. energy of impurity electrons. Defins. of a for the original TiO₇ and for reoxidized partially reduced TiO₈, in 1000 hertzes, gave <math>a_i = 87.5$ and 67.5, resp.; assuming, for a TiO₇ sample reduced 0.5 hrs. in CO, $a_i = 0.0$, and with the exptl. $\Delta E_i = 0.108$ e.v., one finds $m^a/m_a = 16$, i.e., at any rate, the effective m^a is substantially greater than m_a . "Electrons" of such high m^a are more correctly referred to as polarons; in the sense of the theory of Pekar (cf. P. and Delgen, C.4. 43, 37:7d). Values of a caled. with the aid of the extd. m^a agree with the exptl. a only for weakly reduced TiO₈ samples. The observed correlation between A and ΔE , of the type of the Meyer-Neklel (C.A. 33, 2419*) linear relation between log A and ΔB , can be interpreted by the known depressing effect of impurities on a of TiO₅; with increasing A, the dielec. const. a decreases, and, consequently, ΔE (which is inversely proportional to a) increases.

Mar/Apr 52	Metallic Com- nuze, Loningrad 2, pp 155-168	and thermoelec sad SbZn proved ors' exptl data ont with those ond H. Jones.	.s and T.A. Konto-		220789
Semiconductors, Metallic	ties of Some, and V.P. ZkSci USSR Vol XVI, NO	elec Cs3St Autho Treeme	E Properties of Metals and 936) and for MegSn by T.A. 12" 18, 1948).		•
USSR/Physics - Sem	"Semiconductor Properties of Some pounds," B.I. Boltaks, and V.P. Zh Phys-Tech Inst, Acad Sci USSR"Iz Ak Wauk, Ser Fiz" Vol XVI, No	Results of investigations of properties of Mg3Sb2, Mg2Sn, their semiconducting nature. on Mg3St2 and Mg2St are in ag obtained for Mg3Sb2 by N.F. N	("The Theory of the Properties Alloys," Oxforā, 1936) and for rova ("Zhur Tekh Fiz" 18, 1948)	•	.I .a , ZMATHOS
	200 - 13 - 13 - 13 - 13 - 13 - 13 - 13 -				





USSR/Physics - Semiconductors

FD-2846

Card 1/1

Pub. 153-29/30

Author

: Boltaks, B.

Title

: Diffusion of Lead and Antimony in Semiconductive Alloys Bi₂Se₃ and Bi2Te3

Periodical

: Zhur. Tekh. Fiz, 25, 767-768, 1955

Abstract

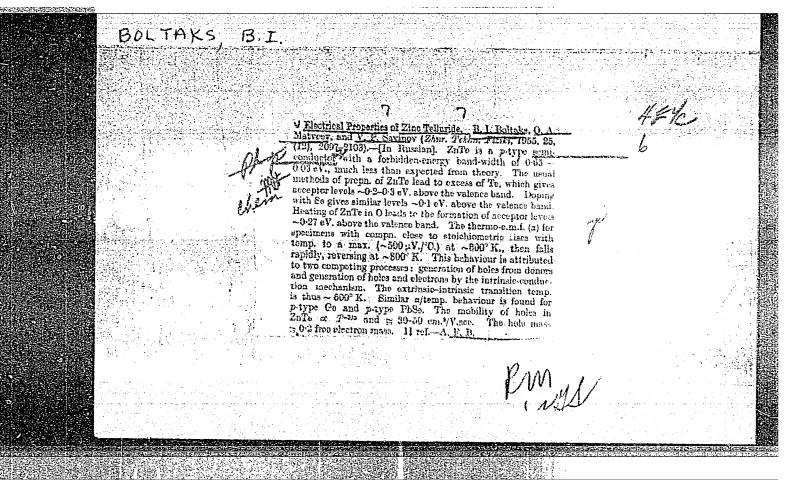
: Electric properties of bismuth selenide and telluride were recently studied by P. Konorov at the Institute of Semiconductors, Acad Sci USSR. Because of the importance of these materials, the diffusion of lead and of antimony in these semiconductors were studied and experimental data on the diffusion coefficient are presented in

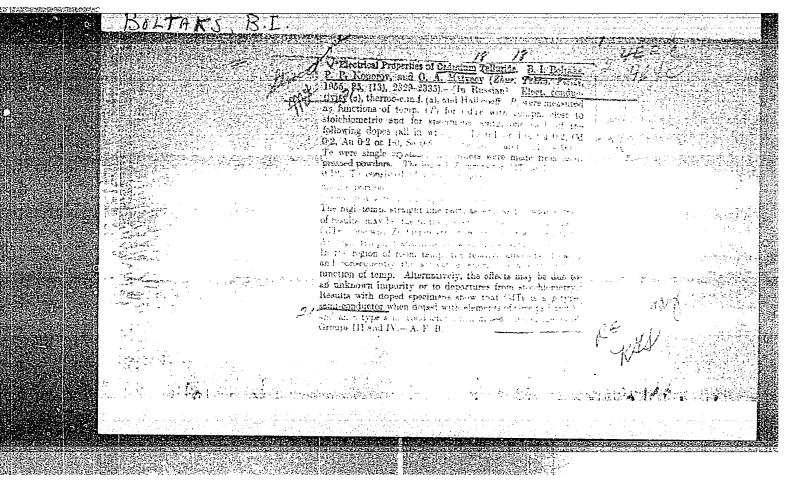
graphs. Two US and two USSR references.

Institution

Submitted

: January 28, 1955





BOLTAKS, B.I.

USER/Physice

Card 1/1

Pub. 22 - 16/49

Authors

s Bolteks, B. I.

Title

About the diffusion of antimenium and tin in semi-conductive compound SbZn

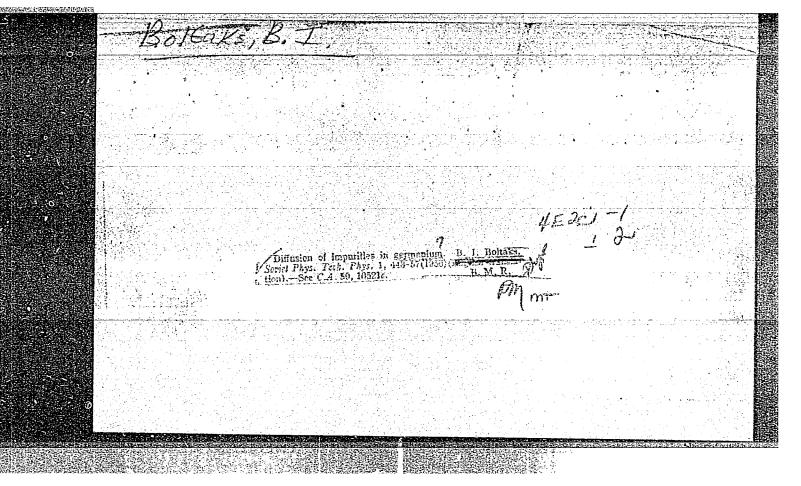
Periodical : Dok. AN SSCR 100/5, 901-903, Feb 11, 1955

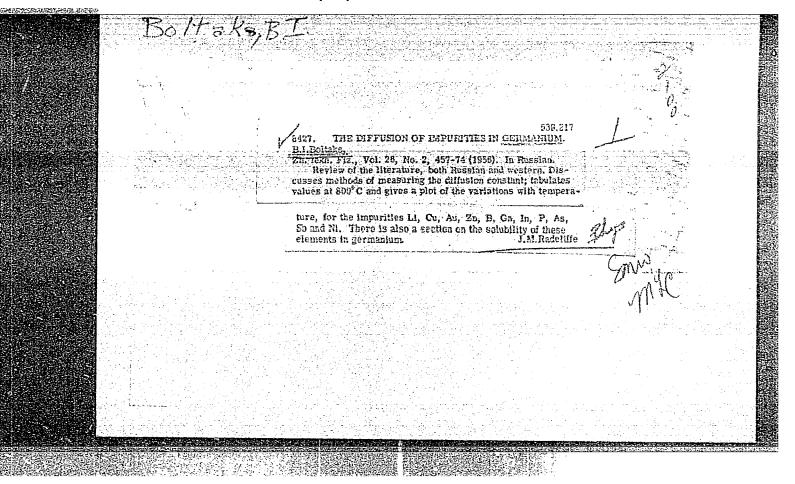
Abstract

Experiments with Sb and Sn are described. The experiments were conducted for the purpose of learning the dependence of diffusion coefficients of the mentioned elements in semi-conductive compounds, such as SbZn, on the temperature. Measurements of the diffusion coefficients were made by the usual method of succesively taking off layers, using for an indicator of radioactive isotopes, in the experiments mentioned, the Sb¹²⁴ and Sn¹¹³ isotopes. Eight references: 7 USA and 1 USSR (1947-1954). Diagram.

Institution : Academy of Sciences of the USSR, laboratory of semi-conductors

Presented by : Academician A. F. Ioffe, October 9, 1954





BOLTAKS, B.I.

SUBJECT

USSR / PHYSICS

CARD 1 / 2

PA - 1673

AUTHOR

BOLTAKS, B. I., MOCHOV, JU. N.

TITLE PERIODICAL The Diffusion of Lead in Lead Telluride. Zurn.techn.fis, 26, fasc.11, 2448-2450 (1956)

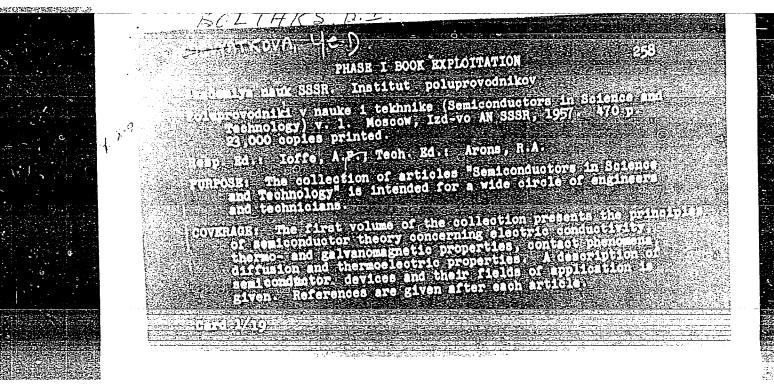
Issued: 10 / 1956

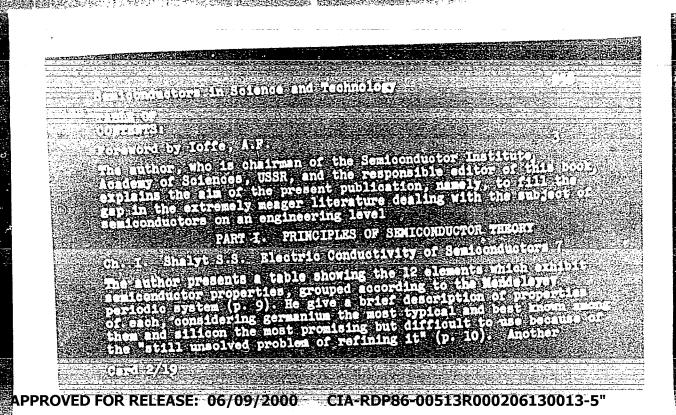
The investigation was carried out on monocrystalline castings of PbTe which had been produced from sublimed tellurium and the purest lead. The monocrystals were produced according to the method developed by BRIDGMAN by means of a device which was similar to that constructed by W.D.LAWSON, J.Appl. Phys. 22, 12, 1444 (1951). All PbTe-monocrystals produced here had hole-like conductivity. The samples which were cut out of the then annealed castings were carefully ground so as to be plane-parallel. A layer of lead with a thickness of from 3 to 5 micron was steamed on to one of the samples in the vacuum, whereupon the samples were subjected to a process of diffusion annealing in evacuated and soldered ampules at a temperature of from 250 to 500° C. Annealing took place in electric ovens with thermoregulators. In the course of diffusion annealing two samples, the surfaces of which (which were coated with layers of lead) touched each other, were annealed together in one ampule in each case. During heat treatment control samples were heated in a separate ampule but at the same temperature for control purposes, on which, after annealing, the electric conductivity and HALL'S constant were measured.

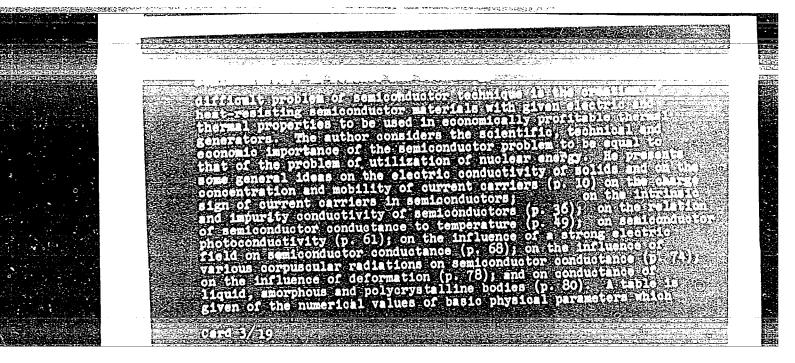
After these provisional results the introduction of lead into the lead telluride changes its hole-like conductivity into electronic conductivity. Therefore

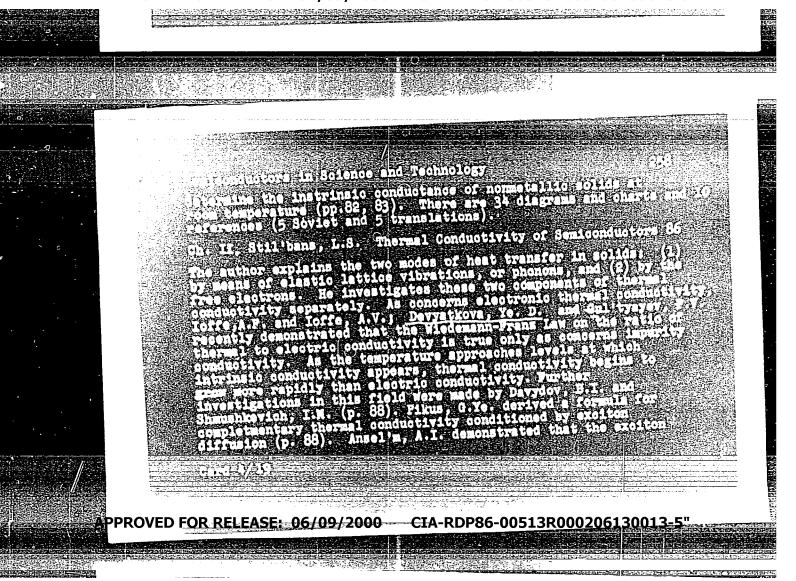
Zurn.techn.fis, 26, fasc. 11, 2448-2450 (1956) CARD 2 / 2 the diffusion coefficients of the lead in the PbTe were measured by the method of electron-hole transitions. The microscopic system used for purposes of observation made the immediate determination of the electron-hole transition possible. A formula for the computation of the diffusion coefficients is given. A table contains the data on the position of the electron-hole transitions, the temperature, the duration of the diffusion annealing, the concentration of the acceptor admixtures in the samples investigated here, and on the diffusion coefficients computed by means of the formula mentioned here. According to these data the diffusion coefficient in PbTe changes in accordance with the following exponential law: $D = 2,9.10^{-5}$ exp (-0,6/KT) cm²/sec. The low activation energy (0,6 eV) and the high diffusion velocity at temperatures which are comparatively low (compared to the melting temperature of PbTe, namely 905° C) leads us to suppose that diffusion is here realized by the shifting of positively charged lead ions in the interspaces between the nodes (like in the case of the diffusion of copper in germanium). Further investigations of this diffusion as well as of other phenomena will supply more detailed information concerning the semiconductor properties of PbTe.

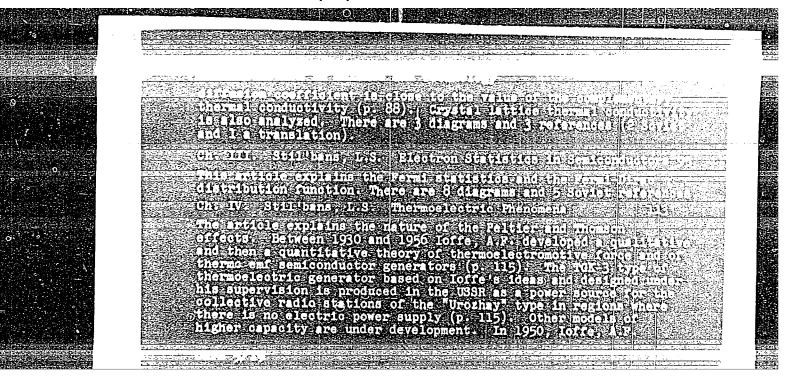
INSTITUTION: Institute for Semiconductors of the Academy of Science in the USSR

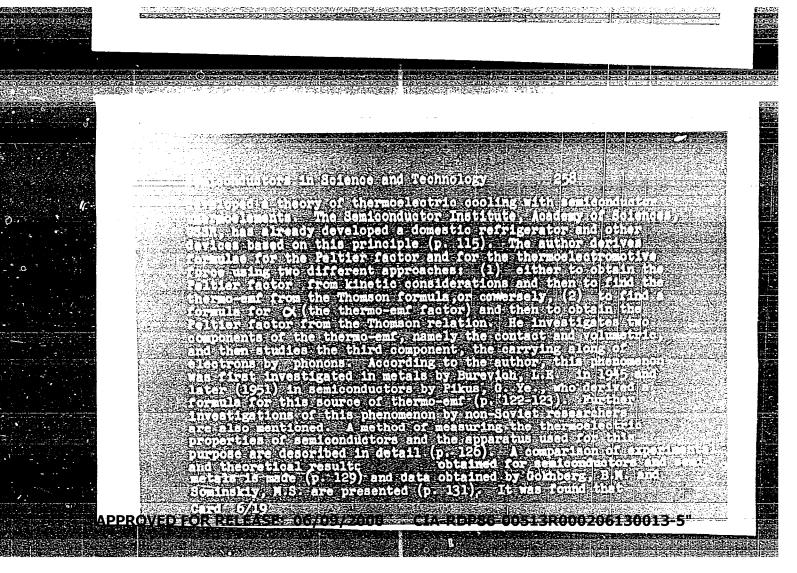


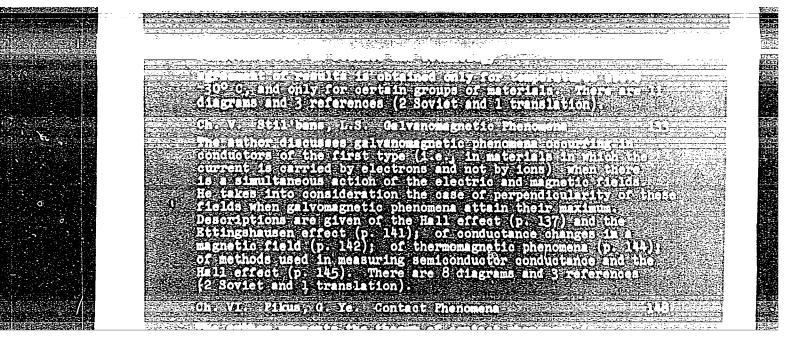












```
in 10 km s mid h chimales

and explains the principles of operation of

(inclient tensistors) high requency p-n-p and an-p-n

(inclient tensistors) and tensistors (the life)

(inclient tensistors) and tensistors (the life)

(inclient (vector) tensistors and 20 in shelling

(inclient (vector) tensistors and 20 in shelling

(inclient (vector) tensistors and 20 in shelling

(inclient state discussed (i)) Be in weenfalling

(inclient subjects as discussed (i)) Be in weenfalling

(i) this personation results on the inclient of

(i) this personation results on the inclient of

(ii) this personation results on the inclient of

(ii) this personation results on the inclient of

(iii) the inclient of the inclient of the inclient of

(iii) the inclient of the inclient of the inclient of

(iii) the inclient of the inclient of the inclient of

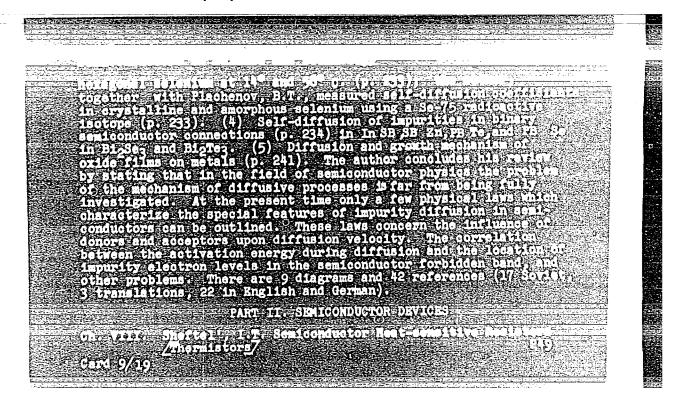
(iii) the inclient of the inclient of the inclient of

(iii) the inclient of the inclient of the inclient of

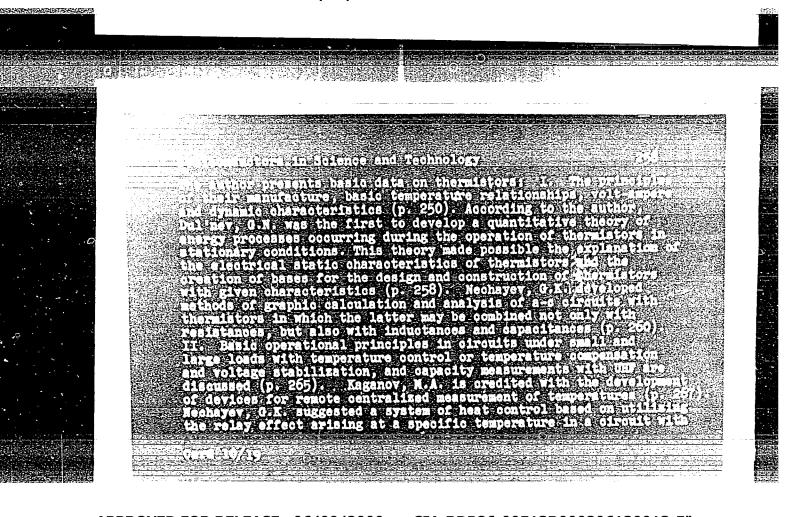
(iii) the inclient of the inclient of
```

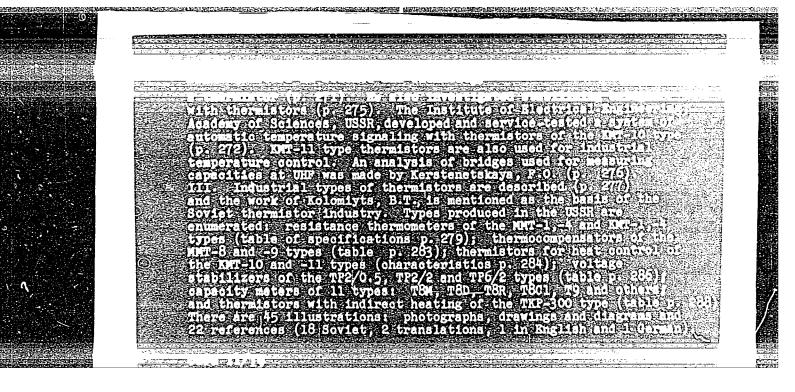
CIA-RDP86-00513R000206130013-5"

PPROVED FOR RELEASE: 06/09/2000



"APPROVED FOR RELEASE: 06/09/2000 CIA-RDP86-00513R000206130013-5

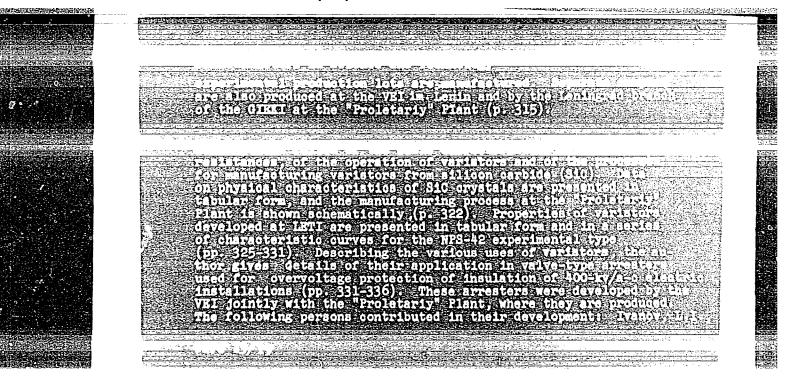


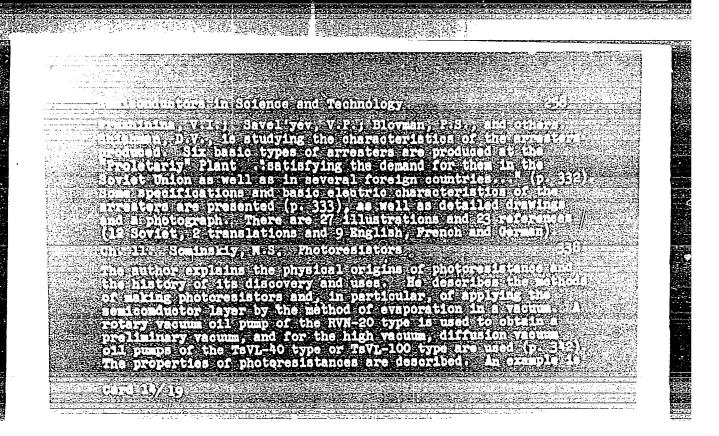


```
IN SOLENGE and Yearnology

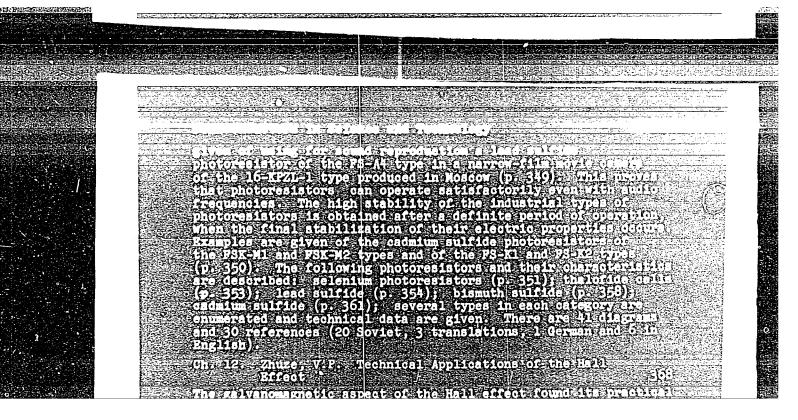
[ICALT, D. If Semiconductor Boloscient

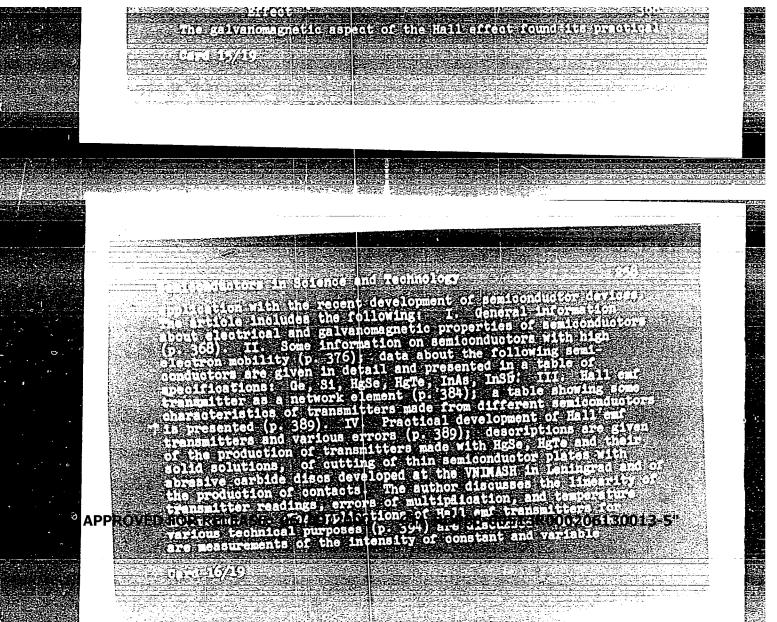
(ICALT, D. If Semic
```

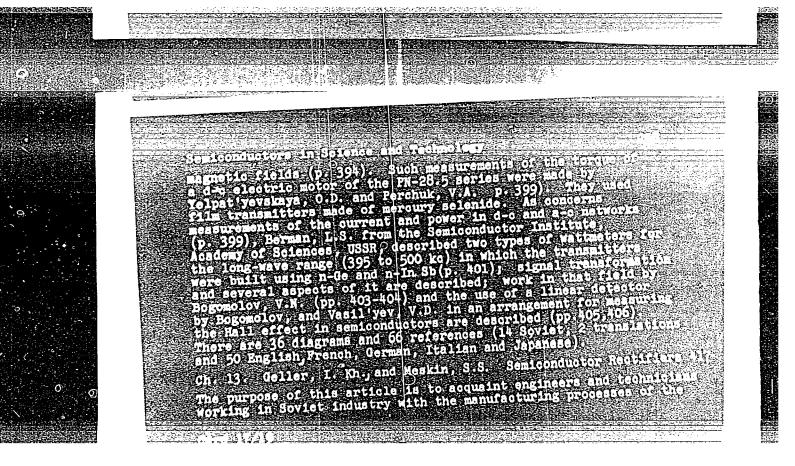


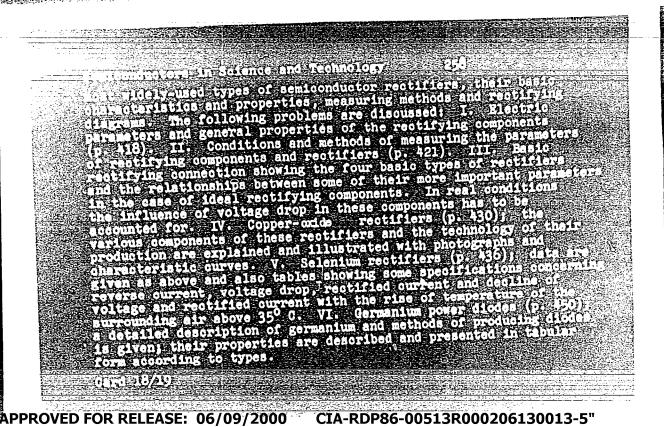


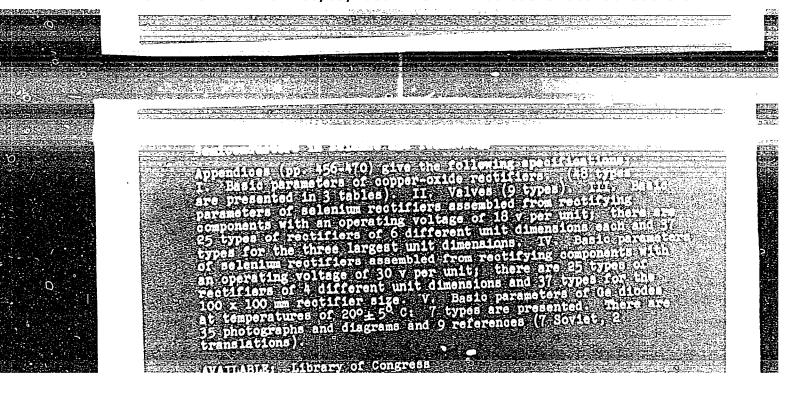
"APPROVED FOR RELEASE: 06/09/2000 CIA-RDP86-00513R000206130013-5











SOV/137-59-1-986

Translation from: Referativnyy zhurnal. Metallurgiya, 1959, Nr 1, p 134 (USSR)

AUTHOR: Boltaks, B. I.

TITLE: On the Diffusion Coefficients of Some Impurities in Germanium (O koeffitsiyentakh diffuzii nekotorykh primesey v germanii)

PERIODICAL: V sb.: Vopr. metallurgii i fiz. poluprovodnikov. Moscow, AN SSSR, 1957, pp 121-129

ABSTRACT: The author measured the diffusion coefficients D of Sb, Pb, and Sn in single-crystal specimens of pure Ge in the 600-900°C range.

The D values obtained for Sb satisfy the equation D=0.6 exp(-2.2 ev/kT).

The diffusion of Sn in "n" type Ge obeys the equation:

D=1.7 · 10-2 exp (-1.9 ev/kT). A sharp increase in the diffusion of

 $D=1.7 \cdot 10^{-2}$ exp (-1.9 eV/kT). A sharp increase in the diffusion of of Sn was observed when the low concentrations (~ 10^{-16} cm⁻³) of

impurities were increased.

A. R.

CIA-RDP86-00513R000206130013-5 "APPROVED FOR RELEASE: 06/09/2000

(D.) BOLTAK

USSR / PHYSICS SUBJECT

CARD

PA - 1950

AUTHOR

BOLTAKS, V.I., KULIKOV, G.S.

TITLE

On the Diffusion of Indium Antimony and Tellurium in Indium-

Antimonide.

PERIODICAL

Zurn.techn.fis, 27, fasc.1, 82-84 (1957)

Issued: 2 / 1957

The present work contains experimental data concerning the coefficients of this diffusion. The authors carried out these investigations on monolithic coarsegrained bars of InSb, which were produced from chemically pure components. The diffusion coefficients were measured by the usual method of the successive removal of layers and by using the radioactive isotopes In-114, Sb-124 and Te-127 as indicators. Diffusion-annealing was carried out at various temperatures within the interval of 300 - 500°C; the duration of this annealing varied according to temperatures from 50 to 180 hours.

A diagram illustrates the dependence lg D = f(1/T) obtained at various temperatures of diffusion annealing. Here D denotes the diffusion coefficient and T the temperature of annealing in K. Thus it was found that the coefficients of the diffusion of indium, antimony and tellurium change in InSb crystals on the occasion of the change of temperature according to the usual exponential law. The corresponding values of the most important diffusion parameters are given. These diffusion coefficients decrease at one and the same temperature of the diffusion annealing on the occasion of transition from In to Te and further to Sb. With increasing temperature this difference decreases. The extrapolation of the data ob-

Žurn.techn.fis, 27, fasc.1, 82-84 (1957)

CARD 2 / 2

PA - 1950

tained into the domain of higher temperatures shows that near the melting temperature of InSb the diffusion coefficients of In, Te and Sb have one and the same value. This is obviously a consequency of the sharp decrease of the amount of the potential barriers between the nodes of the crystal lattice near melting temperature. The differences of activation energy in connection with the diffusion of In, Te and Sb in InSb are closely connected with the mechanism of the influence exercised by these elements upon the electric properties of InSb. The introduction of an admixture of tellurium into InSb turns the hole-like conductivity into an electronic conductivity, and this change is due to the replacement of pentavalent antimony in the tetrahedron-like covalent binding of InSb by hexavalent tellurium. Like antimony, tellurium must diffuse along the vacant nodes.

The interpretation of the data concerning the diffusion of In in InSb is more complicated. Indium apparently does not diffuse in InSb along the vacant nodes, but along the domains located between the nodes in form of neutral atoms of positive ions. The influence exercised by the surplus of indium upon the character of the conductivity of InSb still remains to be investigated.

INSTITUTION: Institute for Semiconductors of the Academy of Science in the Ussr, Leningrad

BoltAKS, B.J.

AUTHORS:

Boltaks, B. I., and Plachenov, B. T.

57-10-4/33

TITLE:

A Note on Autodiffusion in Selenium (Samodiffuziya v selene).

PERIODICAL:

Zhurnal Tekhn. Fiz., 1957, Vol. 27, Nr lo, pp. 2229-2231 (USSR).

ABSTRACT:

Data are given for the autodiffusion coefficient of amorphous and crystalline selenium. They were obtained on the basis of experiments on "selenium for rectifiers" beeing the purest obtainable material. It is shown, that these coefficients are essentially different in the two cases and that this difference increases with a temperature rise. In the case of amorphous selenium the modification of the autodiffusion coefficient with temperature is characterized by the greater value of the factor D in front of the exponential function and the

greater value of the activation energy (As). In the case of crystalline selen the inverse process is observed. The marked increase of the factor in front of the exponential function at the transition from crystalline to amorphous selenium is connected with an increase of the entropy accompanying this process. The difference of the activation energy is apparently caused by the different kind of processes of autodiffusion in crystalline and amorphous selenium.

Card 1/2

There are 1 table, 1 figure and 1 Slavic reference.

A Note on Autodiffusion in Selenium.

57-lo-4/33

ASSOCIATION:

Institute for Semiconductors AN USSR, Leningrad (Institut policy

provodnikov AN SSSR., Leningrad).

SUBMITTED:

March 4, 1957.

AVAILABLE:

Library of Congress.

Card 2/2

BOLTAKSBI

AUTHORS:

Boltaks, B. I., Tszyan Pi-khuan,

57-1-30/30

TITLE:

On the Change of the Coefficient of the Thermoelectromotive Power of Tellurium with Copper and Antimony Admixtures on the Occasion of Transition from the Solid into the Liquid State (Izmeneniye koeffitsiyenta termoeds tellura s primesyami medi i sur'my pri perekhode ot tverdogo k zhidkomu sostoyaniyu)

PERIODICAL:

Zhurnal Tekhnicheskoy Fiziki, 1958, Vol. 28, Nr 1, pp. 222-224 (USSR)

ABSTRACT:

The experimental data obtained are given. Tellurium whichwas sublimated twice and which showed a hole - conductivity of ~2-3 Ohm-1.cm-1 and a coefficient of thermoelectromotive power of ~400 aV/degree was used as basic material on the occasion of the investigation of the thermoelectric properties of tellurium with copper admixtures. Hole-tellurium parified by means of zone recrystallization served as basic material for the investigation of the thermoelectric properties of tellurium with antimony admixtures. It can be seen from the data obtained that the coefficient of the thermoelectromotive power in the case of liquid tellurium has positive sign as to "allumel" (alloy for thermoelements, main material nickel with Al, Mn, Si, Co) and amounts to ~40-50 aV/degrees. It does not depend on the rate of admixtures in the bar. Thermo-

Card 1/3

On the Change of the Coefficient of the Thermoelectromotive Power 57-1-30/30 of Tellurium With Copper and Antimony Admittures on the Coasion of Transition from the Solid into the Liquid State.

electromotive power of solid tellurium (near the melting point) depends on the presence of the admintures. Copper admixtures considerably increase the coefficient of the thermoelectromotive power with the initial tellurium which fact is probably in connection with its donator properties in tellurium. In the range of \sim 50-2000C thermoelecromotive power changes its signs and bacomes electronic. An admixture of anitmony shows less effects on the coefficient of the thermoelectromovite power of tellurium but also in this case the same tendency towards the increase of the coefficient of thermoelectromotive power in solid state near the melting point can be observed. Between solid and liquid tellurium the coerficient of thermoelectromotive power increases with the admixture of copper and antimony. Thus, the coefficient of the thermoelectromotive power between solid and liquid tellurium can be changed by means of different admixtures. There are 3 figures, 2 tables and 4 Slavic references, 2 of which are Slavic.

Card 2/3

On the Change of the Goefficient of the Thermoelectromotive Power 57-1-30/30of Tellurium With Copper and Antimony Admixtures and of the ccasion of Transition from the Solid into the Liquid State.

ASSOCIATION: Leningrad Polytechnical Institute imeni M.I. Kalinin (Leningradskiy

politechnicheskiy institut im. H.I. Kalinina)

SUBMITTED: March 30, 1957

AVAILABLE: Library of Congress

Card 3/3

AUTHORS:

Boltaks, B. I., Sozinov, I. I.

57-28-3-33/33

TITLE:

On the Diffusion of Copper in Silicon

(O diffuzii medi v kremnii)

PERIODICAL:

Zhurnal Tekhnicheskoy Fiziki, 1958, Vol. 28, Nr 3,

pp. 679-679 (USSR)

ABSTRACT:

In a short report the authors here give the results obtained from a measurement of the coefficients of the diffusion of copper in silicon for the range of from 800 - 1100°C. The measurements were performed according to the usual method

of a successive removal of layers with the use of the radioactive isotope of copper - 64 as an indicator. According to the obtained data the diffusion coefficient varies from

 4.10^{-7} cm²/sec (at 807° C) to 4.10^{-6} cm²/sec (at 1094° C). The modification of the diffusion coefficient with temperature can be expressed in the following way: $D_{\text{Cu}\rightarrow\text{Si}}=4.10^{-2}$ exp

(-1,0 eV/kT)cm²/sec. Thus copper in the range of from 800 to 1100°C possesses an anomalously high rate of diffusion in silicon. In comparison with germanium, however the

Card 1/2

On the Diffusion of Copper in Silicon

57-28-3-33/33

diffusion coefficients at the same temperature are here 15-to 2c-fold less. Such a difference is entirely justified in connection with the more solid linkage and the correspondingly much higher melting point of silicon as compared to germanium. At 875°C no influence of the constant electric field upon the diffusion of copper in silicon could be found to exist. There are 1 figure and 2 references.

ASSOCIATION:

Institut poluprovodnikov AN SSSR, Leningrad

(Leningrad Institute for Semiconductors, AS USSR)

SUBMITTED:

August 17, 1957

1. Copper--Diffusion 2. Silicon--Analysis 3. Diffusion--Tempera-

ture factors 4. Diffusion-Electrical factors 5. Copper

isotopes (Radioactive)--Applications

Card 2/2

USCOMM-DC-60232